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## Spontaneous emission and collection efficiency enhancement of single emitters in diamond via plasmonic cavities and gratings

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We demonstrate an approach, based on plasmonic apertures and gratings, to enhance the radiative decay rate of single nitrogen-vacancy (NV) centers in diamond while simultaneously improving their collection efficiency. Our structures are based on metallic resonators formed by surrounding sub-wavelength diamond nanoposts with a silver film, which can enhance the spontaneous emission rate of an embedded NV center. However, the collection efficiency of emitted photons remains low due to losses to surface plasmons and reflections at the diamond-air interface. In this work, we mitigate photon losses into these channels by incorporating grating structures into the plasmonic cavity system. © 2013 AIP Publishing LLC. [http://dx.doi.org/10.1063/1.4817397]

Diamond is a promising material for integrated quantum photonics due to its excellent material properties and the optical response of its defect centers, which can include stable single photon emission at room temperature. In particular, the nitrogen-vacancy (NV) center provides optical addressability and readout of electron and nuclear spin states that are associated with long coherence times,<sup>1</sup> which make it ideal for applications such as quantum information processing, quantum communication,<sup>2</sup> and magnetometry.<sup>3</sup> For these applications, it is advantageous to embed color centers into diamond-based nanophotonic devices to control the efficiencies of single photon production and collection.<sup>4</sup>

We have previously presented the "diamond-silver aperture," consisting of a diamond nanopost (of radius  $r_d \approx 50 \,\mathrm{nm}$  and height  $h_d \approx 220 \,\mathrm{nm}$ ) covered in silver (Ag),<sup>5,6</sup> which supports guided modes that become more tightly confined in the dielectric with decreasing  $r_d$ . Reflections between the facets of the aperture form resonances that can be tailored by changing  $r_d$  and  $h_d$  and used to modify the spontaneous emission rate (SE) of an interacting NV center in the nanopost. The hybrid structure can be implemented by a combination of ion implantation and topdown nanofabrication, and the coupling between the emitter and resonator has yielded a six-fold Purcell enhancement of the NV center SE compared to emission in the bulk.<sup>6</sup> While the fluorescence intensities from NV centers in the diamond-Ag apertures have increased from those in the bulk due to faster radiative decay rates, only about 3% of the emitted photons are collected, due to coupling of the emission to surface plasmons and total internal reflection (TIR) at the diamond-air interface. Therefore, it is desirable to engineer the directionality of the emission via periodic corrugations in the metal film<sup>7,8</sup> which coherently scatter surface plasmons, thereby increasing the fraction of optical power going into the 25° critical cone set by TIR. In this work, we experimentally demonstrate a platform to integrate the apertures with plasmonic gratings to improve the collection efficiency of the emitted single photons from cavity-enhanced NV centers.

Similar approaches to provide more efficient extraction of emitted light and to manipulate the angular pattern of the radiation have been proposed<sup>9</sup> and demonstrated with quantum dots,<sup>10-12</sup> fluorescent molecules,<sup>13,14</sup> and NV centers in diamond nanocrystals.<sup>15,16</sup> The geometry presented here is based on emitters in the bulk crystal, thus avoiding the placement techniques generally associated with nanocrystal platforms. We add concentric metallic rings around the diamond-Ag nano-cavity such that the final structure (Fig. 1(a)) consists of periodic grooves in the Ag film filled with a dielectric material around a central aperture. For the dielectric filler, we have selected silicon dioxide  $(SiO_2)$ rather than the more obvious choice of diamond so that the gratings can be controlled independently from the diamond nanoposts. Additionally, since our sample fabrication involves using blanket ion implantation to generate a layer of NV centers, molding the grooves with SiO2 can avoid contribution to the single photon emission by extraneous fluorescence from embedded emitters in the gratings.

We performed 3D finite-difference-time-domain (FDTD) simulations to model our structures. The diamond aperture has  $r_d$  between 50 nm and 60 nm and  $h_d = 220$  nm, with a radially polarized dipole optimally placed in the field maximum of the cavity. The Purcell factor,<sup>17</sup> which is the SE rate enhancement experienced by the dipole in cavity, is calculated<sup>5</sup> and shown in Fig. 1(b). The resonant wavelengths range from 640 nm to 680 nm, which overlap well with the zero phonon line (637 nm) and the peak emission (around 670 nm) of the NV center fluorescence, respectively.

The general grating design is similar to that described in Ref. 8 and consists of five grooves, each with the same width w = 80 nm and height  $h_g = 100$  nm. The distance between the first groove and center of the structure is a = 350 nm, and the periodicity of the subsequent grooves is b = 280 nm. These parameters were selected to overlap the resonant response of the gratings with that of the center aperture, and it can be seen in Fig. 1(b) that the addition of grating has led to an increase in the SE enhancement when the resonant responses are matched, while otherwise the grating induces

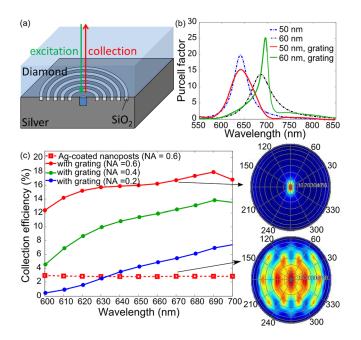


FIG. 1. (a) Schematic of the diamond-Ag grating structure. (b) Calculated Purcell factor for Ag-coated nanoposts with and without grating. (c) Plot of collection efficiency as a function of wavelength for the non-grating and grating cases under different collection optics. Our microscope objective has a N.A. of 0.6. Insets: simulated angular distributions in diamond for the power emitted from (top) a diamond-Ag aperture surrounded by grating and (bottom) the same-radius diamond-Ag aperture without grating at a wavelength of 670 nm.

only small shifts in the cavity resonance wavelength for the range of post radii we studied. The interference between the cavity and grating resonances has also led to a Fano lineshape. The addition of the plasmonic corrugations increases the overall collection efficiency up to 16% over a broad range of wavelengths around the resonance, which represents a 5-fold improvement over the bare aperture case (around 3%). This is shown in Fig. 1(c) and was calculated by projecting the electric fields obtained above the device into the farfield, applying the Fresnel coefficients and Snell's law<sup>18</sup> to take into account the diamond-air interface, and finally dividing the integrated power emitted into the 37° cone set by our collection objective (with numerical aperture N.A. of 0.6) by the total power emitted by the dipole. We have also plotted the collection efficiency under collection with different N.A.. An efficiency of around 10% is still possible with a N.A. of 0.2–0.4, suggesting that the structure might be compatible with direct collection by optical fibers. The improvement in collection efficiency is accompanied by modification of the far field profile of the radiation, in which emission is strongly concentrated at small angles (Fig. 1(c)).

We fabricated arrays of diamond nanoposts of radii from 50 nm to 70 nm and height around 240 nm in an electronic grade diamond that was implanted with a shallow layer of NV centers 90 nm below the surface, using electron beam lithography and reactive ion etching in an  $O_2$  plasma.<sup>4</sup> To realize the plasmonic corrugations, we defined rings (with grating periods *a* from 350 nm to 360 nm and *b* from 280 nm to 300 nm) of flowable oxide around the fabricated nanoposts in a second electron beam lithography step using a negative electron beam resist (hydrogen silsesquioxane). Scanning electron microscope (SEM) images indicate that the gratings and pre-existing structures are well aligned (Figs. 2(a)-2(c)). The images shown here are taken of a different set of samples than the ones optically characterized; this is to avoid introducing contamination (such as carbon deposition) to the devices during the imaging process that can lead to unwanted fluorescence. To further ensure the cleanliness of the surface, we cleaned the diamond sample in a Piranha solution (3:1 H<sub>2</sub>SO<sub>4</sub>:H<sub>2</sub>O<sub>2</sub>) and annealed the sample at 465 °C in an O<sub>2</sub> purged environment. The final structure was realized by capping device with sputtered Ag.

As mentioned earlier, we characterized the diamond-Ag apertures by exciting the sample with a continuous wave pump laser at 532 nm and collecting the NV center emission (filtered by a bandpass filter from 650-800 nm) through the bulk diamond crystal in a home-built confocal microscope with a 0.6 N.A. microscope objective.<sup>6</sup> The fluorescence was focused onto one end of a single-mode,  $2 \times 2$  fiber coupler, with each output connected to an avalanche photodiode (APD) for photodetection. A confocal scan image of the sample (Fig. 2(d)) from one of the APD outputs shows that emission from posts surrounded by gratings are brighter than those without gratings under the same pump power (around 1 mW). To confirm single photon character from individual NV centers in the structures, the photon arrival times at the APDs were recorded and plotted in a histogram of coincidence counts as a function of time delay. After normalization, we obtain the second order autocorrelation function,  $g^{(2)}(\tau)$  (Fig. 3(a)). For the two devices indicated with arrows in Fig. 2(d),  $g^{(2)}(0) < 0.5$ , demonstrating that they each contain a single emitter. These nanoposts have the same post radius of 55 nm, so that their resonator characteristics should be very similar (as can be seen by the background-subtracted photoluminescence spectra shown in Fig. 3(b)), thus allowing us to compare the effect of the grating.

The fluorescence intensities, *I*, of the nanoposts were measured under different pump powers *P* (Fig. 3(c)). The background contribution to the fluorescence was inferred directly from  $g^{(2)}$  measured at various *P*.<sup>19,20</sup> The background subtracted count rate is the NV fluorescence intensity and is fitted to the saturation model,  $I(P) = \frac{I_{sat}}{1+P_{sat}/P}$ , where  $I_{sat}$  and  $P_{sat}$  are the saturation intensity and power, respectively.  $I_{sat}$  of the post with grating is 704 ± 38 kcps while that for the bare cavity is 237 ± 17 kcps.  $P_{sat}$  are comparable in the two cases.

Contributions to increased Isat can be attributed to improvement in collection efficiency and/or Purcell enhancement. The latter can be inferred from the emitter lifetime, which was measured using excitation by green pulses at a 76 MHz repetition rate (Fig. 3(d)). The fluorescence intensity time traces were fitted to an exponential decay model, and the resulting time constants are 3.2 ns and 2.5 ns for the nongrating and grating cases, respectively. The corresponding Purcell factors, calculated by comparing the NV lifetime in cavity with that in the bulk (around 16 ns in 90-nm implanted NV centers), are roughly 6. The discrepancies between the observed Purcell enhancements and the predicted values can be attributed to geometric effects (most notably the tapering in the sidewalls of the nanoposts from the reactive ion etching process), straggle in the implantation depth, emitter spectrum, and misalignment between the polarization angle of

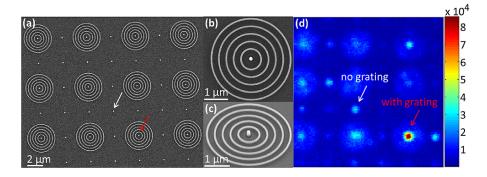


FIG. 2. (a)–(c) SEM (taken before Ag deposition) and (d) confocal scan images (taken after sample has been embedded in Ag) of arrays of diamond nanoposts (white arrow) in which alternating posts are surrounded by  $SiO_2$  rings (red arrow).

the dipole moment and the [100] diamond crystal plane. In particular, we calculated that the peak emission enhancement is reduced by about 40% from the straight sidewall case for a  $85^{\circ}$  sidewall angle,<sup>20</sup> which matches well with experimentally obtained values. Taking Purcell enhancements into account, the improvement in collection efficiency provided by the grating is around 2.3.

Finally, we consider the effect of pinhole filtering on the overall collection efficiency of the setup. While the footprint of our grating devices spans over 5  $\mu$ m, the measured spot size on our confocal microscope has a full-width half-maximum of 610 nm. Therefore, some of the light scattered by the grating is not collected by our confocal system. To increase the area of collection, we replaced the single-mode fiber (SMF), which has a core size of around  $4.5 \,\mu m$ , with a multi-mode fiber (MMF) with a core size of  $62.5 \,\mu\text{m}$ . Since the role of the pinhole is to reduce the area and depth of collection, enlarging it would increase the area being probed in a way that is proportional to the increase in its size. However, we also sacrificed the filtering of unfocused light which led to an overall increase in background coming from the larger volume of diamond being probed. Nonetheless, as shown in Figs. 4(a) and 4(b), the scan images taken with SMF and MMF show similar features, with increased collected count rates by several fold in the latter. Due to the higher background,  $g^{(2)}(0)$  is slightly above 0.5 in the MMF case (Fig. 4(c)). To quantify the intensity increase in a grating device with confirmed single photon character, we compared its saturation behaviors under the two

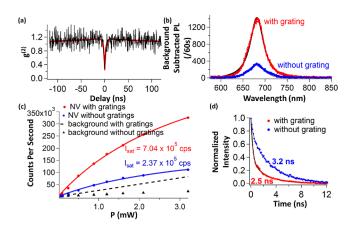


FIG. 3. (a)  $g^{(2)}$  plot of a diamond-Ag nanopost with grating. The black line indicates raw data, while the red line is the fit. Comparison of (b) photoluminescence (PL) spectra, (c) saturation intensities (the background-subtracted fluorescence, and the background count rates are shown), and (d) fluorescence time decay traces of the grating and non-grating cases. The solid lines in (d) indicate fits to an exponential model.

collection channels (Fig. 4(d)). After background subtraction,  $I_{sat}$  is almost 4 times higher than that collected by a SMF (from 337 kcps to 1.35 Mcps), while  $P_{sat}$  is mostly unchanged between the two measurements. Photon count rates of the structures under MMF collection are thus comparable to those obtained from nanowires (~500 kcps)<sup>21</sup> and solid-immersion lenses (~2 Mcps).<sup>22</sup> Ultimately, to optimize both the collection of all scattered signal from the grating and spatial filtering of the background, a variable pinhole may be used.

We have experimentally demonstrated the integration of plasmonic gratings with single NV centers in diamond-Ag apertures. With some improvements mentioned below, our system is potentially promising for shaping and collimating single photon beams on chip. The observed improvement in collection efficiency is modest (a factor of 2.3) and is about half of the predicted value. This could be attributed to deviation of the dipole from the field maximum of the cavity, which affects the coupling of emission to surface plasmons. Additionally, the slight taper in the sidewall degrades the confinement and further reduces the coupling to propagating surface plasmons.

Overall, the device yield is poor, with only about 4 working grating devices (qualified by having  $g^{(2)}(0) < 0.5$ ) out of over 100 tested across several experimental attempts. These devices have saturation intensities from 337 to 704 kcps, which are consistently higher those from Ag-coated

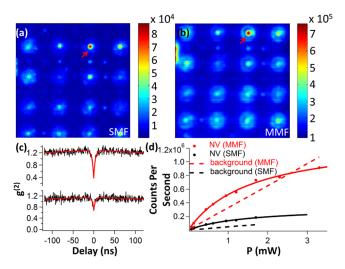


FIG. 4. Confocal scans of the same area of the sample as imaged using a (a) SMF and (b) MMF at the same power. (c)  $g^{(2)}$  plot of the device (indicated by the red arrows) under collection by a SMF (top) and MMF (bottom), taken at comparable powers. (d) Saturation curves under collection by a SMF and MMF.  $I_{sat}$  are  $3.37 \times 10^5$  cps and  $1.35 \times 10^6$  cps, respectively.  $P_{sat}$  are around 1.5 mW for both fits.

nanoposts. To achieve true scalability of the system, certain material issues need to be resolved. These include the reliable deposition of high-quality Ag films and better understanding of the chemical and electronic interactions between diamond and Ag at the interface.

The collection efficiency in our system could be further improved by the addition of an anti-reflection (AR) coating, the simplest implementation of which is a 130-nm thick layer of SiO<sub>2</sub> that can enhance transmissivity through the diamond surface by about 20% around the peak emission of the NV center.<sup>20,23</sup> Finally, optimized simulations indicate that a collection efficiency of up to 64% as well as other beam designs are attainable.<sup>20</sup>

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- <sup>1</sup>F. Jelezko and J. Wrachtrup, Phys. Status Solidi A **203**(13), 3207–3225 (2006).
- <sup>2</sup>L. Childress and R. Hanson, MRS Bull. 38(2), 134–138 (2013).
- <sup>3</sup>S. Hong, M. S. Grinolds, S. Michael, L. M. Pham, D. Le Sage, L. Luan,
- R. L. Walsworth, and A. Yacoby, MRS Bull. 38(2), 155–161 (2013).
- <sup>4</sup>B. J. M. Hausmann, J. T. Choy, T. M. Babinec, B. J. Shields, I. Bulu, M. D. Lukin, and M. Loncar, Phys. Status Solidi A 209, 1619–1630 (2012).

- <sup>5</sup>I. Bulu, T. Babinec, B. Hausmann, J. Choy, and M. Loncar, Opt. Express **19**, 5268–5276 (2011).
- <sup>6</sup>J. Choy, B. Hausmann, T. Babinec, I. Bulu, M. Khan, P. Maletinsky, A. Yacoby, and M. Loncar, Nature Photon. **5**, 738–743 (2011).
- <sup>7</sup>H. Lezec, A. Degiron, E. Devaux, R. Linke, L. Martin-Moreno, F. Garcia-Vidal, and T. Ebbesen, Science **297**, 820–822 (2002).
- <sup>8</sup>S. Carretero-Palacios, O. Mahboub, F. Garcia-Vidal, L. Martin-Moreno, S. Rodrigo, C. Genet, and T. Ebbesen, Opt. Express **19**, 10429–10442 (2011).
- <sup>9</sup>I. S. Maksymov, M. Besbes, J. P. Hugonin, J. Yang, A. Beveratos, I. Sagnes, I. Robert-Philip, and P. Lalanne, Phys. Rev. Lett. **105**, 180502 (2010).
- <sup>10</sup>A. G. Curto, G. Volpe, T. H. Taminiau, M. P. Kreuzer, R. Quidant, and N. F. van Hulst, Science **329**, 930–933 (2010).
- <sup>11</sup>N. Livneh, A. Strauss, I. Schwarz, I. Rosenberg, A. Zimran, S. Yochelis, G. Chen, U. Banin, Y. Paltiel, and R. Rapaport, Nano Lett. **11**, 1630–1635 (2011).
- <sup>12</sup>Y. C. Jun, K. C. Y. Huang, and M. L. Brongersma, Nat. Commun. 2, 283 (2011).
- <sup>13</sup>T. H. Taminiau, F. D. Stefani, F. B. Segerink, and N. F. van Hulst, Nat. Photonics 2, 234–237 (2008).
- <sup>14</sup>H. Aouani, O. Mahboub, N. Bonod, E. Devaux, E. Popov, H. Rigneault, T. W. Ebbesen, and J. Wenger, Nano Lett. **11**, 637–644 (2011).
- <sup>15</sup>J. Wolters, G. Kewes, A. W. Schell, N. Nusse, M. Schoengen, B. Lochel, T. Hanke, R. Bratschitsch, A. Leitenstorfer, T. Aichele, and O. Benson, Phys. Status Solidi B 249, 918–924 (2012).
- <sup>16</sup>O. Mollet, S. Huant, G. Dantelle, T. Gacoin, and A. Drezet, Phys. Rev. B 86, 045401 (2012).
- <sup>17</sup>E. M. Purcell, Phys. Rev. 69, 681 (1946).
- <sup>18</sup>E. Hecht, Optics, 4th ed. (Addison Wesley Publishing Company, San Francisco, 2002).
- <sup>19</sup>A. Beveratos, R. Brouri, T. Gagoin, J.-P. Poizat, and P. Grangier, Phys. Rev. A 64(6), 061802 (2001).
- <sup>20</sup>See supplementary material at http://dx.doi.org/10.1063/1.4817397 for details on the data analysis (background extraction from g(2) measurements), parameters used for the FDTD modeling, modeling results for tapered nanoposts and optimized grating designs, and modeling and experimental data on the anti-reflection coating.

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- <sup>21</sup>T. M. Babinec, B. J. M. Hausmann, M. Khan, Y. Zhang, J. R. Maze, P. R. Hemmer, and M. Loncar, Nat. Nanotechnol. 5(3), 195–199 (2010).
- <sup>22</sup>T. Schroder, F. Gadeke, M. J. Banholzer, and O. Benson, New J. Phys. 13(5), 055017 (2011).
- <sup>23</sup>T. K. Yeung, D. Le Sage, L. M. Pham, P. L. Stanwix, and R. L. Walsworth, Appl. Phys. Lett. **100**, 251111 (2012).